## Properties studies on the P-doping NASICON powders

SHUANG ZHANG, ZHIYONG ZHAO, BAOFU QUAN, BENGANG ZHAO, WEIYOU CHEN Institute of Electronic Science and Engineering, Jilin University, Changchun 130023, People's Republic of China E-mail: shuangzhang1978@yahoo.com

The development of solid electrolytes has been given a boost by the discovery of the NASICON structure [1, 2]. Crystalline NASICON generally possesses a rhombohedral structure, in which the BO<sub>6</sub> octahedron share corners with the PO<sub>4</sub> tetrahedron and the SiO<sub>4</sub> tetrahedron. NASICON possesses 'open' structures in which alkali ions can move with significantly reduced activation barriers and hence there is potential for applications as fast ion conductors [3, 4].

However, a lot of evaporation of phosphorus occurred during the sintering process. The framework was destroyed, so that the electronical conductivity was decreased [5–7]. It is obvious that the content of phosphorus is important for electronical conductivity and  $CO_2$  response of sensors. In this work we increased the conductivity and sensitivity to  $CO_2$  of NASICON through diffusing P into NASICON disks in order to increase the content of P in samples.

NASICON powders were synthesized by the sol-gel method [8, 9]. The pellets were obtained by applying a pressure of about  $14 \times 10^3$  kg·cm<sup>-2</sup> for about 10 min. The doping course was completed in a diffusion system. The (NH<sub>4</sub>)H<sub>2</sub>PO<sub>4</sub> powders used as a source for phosphorus were decomposed at 850 °C. The NASICON pellets were sintered at 550 °C and were doped by incorporating P<sub>2</sub>O<sub>5</sub> and P<sub>2</sub>O<sub>3</sub>.

As shown in Fig. 1, the X-ray diffraction (XRD) patterns of NASICON indicated sharp diffraction peaks signifying a high level of crystallinity. The powders presented better structures and showed small peaks of tetragonal zirconia.

X-ray photoelectronic spectra (XPS) analysis was performed in order to determine the chemical nature of the interfacial layer and the chemical composition of the specimens, as shown in Fig. 2. The binding



Figure 1 XRD spectra of NASICON powders by the sol-gel method.

energies of the Na1s, Zr3d, Si2p, P2p and O1s core level electrons for the  $Na_3Zr_2Si_2PO_{12}$  powders before implanting P and after implanting P were determined from the respective XPS and are listed in Table I.

Slight variations have been observed for the binding energies of the Zr3d, Si2p, P2p, O1s and Na1s core level electrons for both samples. These variations have been attributed to the changes in the next nearest neighbors of the cation [5]. The framework was varied with the P content. The ratio of elements of the un-doped and doped samples can be obtained approximately. It is revealed by XPS that the P content of the doped sample was more than the un-doped sample.

Complex impedance analysis of the electrical data has been carried out to determine the electrical conductivity of the samples before implanting P and after implanting P. The relevant data are listed in Table II. The higher conductivity was shown by the doped sample. It is known that the conductivity of NASICON

TABLE I XPS core level binding energies of various elements in (a) un-doped disk (b) doped disk in NASICON systems, values are in eV

	Na1s	Zr3d	Si2p	P2p	O1s
(a)	1069.6	181.5	100.8	132.7	531.5
(b)	1069.8	181.6	100.5	132.9	531.0

TABLE II Electrical data of the un-doped disk and doped disk

Sample	$\sigma_{303 \text{ K}} \text{ (S cm}^{-1})$	$E_{\rm a}~(0.01~{\rm eV})$
Un-doped	$8.36 \times 10^{-4}$	0.33
Doped	$4.05 \times 10^{-3}$	0.32



Figure 2 XPS spectrum for doped and un-doped NASICON samples.



*Figure 3* The CO<sub>2</sub> concentration dependence of the EMF for the NASICON samples.

decreases with a decrease in the lattice constants because of a decrement of the runnel size for  $Na^+$  migration [10]. It is obvious that the doped powders have the better framework in their structure due to the higher P content.

Fig. 3 shows the relationship between the EMF response and the  $CO_2$  gas concentration for the gas sensors prepared using the samples. For the sensor using the doped powder, the measured sensitivity of 71.5 mV/decade was close to the theoretical value of 74.0 mV/decade based on a two-electron electrochemical reaction [11]. In the case of the sensor using the un-doped powder, the sensitivity was 66.3 mV/decade.

## Acknowledgment

The authors thank NSFC (subject No. 60174033) and JPSTC (subject No. 20010323) for financial support.

## References

- 1. H. Y.-P. HONG, Mater. Res. Bull. 11 (1976) 173.
- M. HOLZILGER, J. MAIER and W. SITRTE, Solid State Ionics 94 (1997) 217.
- G. V. PRAKASH, P. NACHIMUTHU, M. VITHAL and R. JAGANNATHAN, Bull. Mater. Sci. 22 (1999) 121.
- 4. M. CATTI, J. Solid State Chem. 156 (2001) 305.
- 5. B. V. R. CHOWDARI, G. V. SUBBA RAO and G. Y. H. LEE, Solid State Ionics 136–137 (2000) 1067.
- 6. G. ADACHI, N. IMANAKA and H. AONO, *Adv. Mater.* 8 (1996) 127.
- 7. F. JIE, Solid State Ionics 96 (1997) 195.
- B. F. QUAN, S. ZHANG, X. N. LIU, L. H. CHEN and X. L. LIU, J. Trans. Techn. 21 (2002) 11.
- 9. S. ZHANG, B. F. QUAN, Z. Y. ZHAO, Y. C. ZHOU, Y. H. HE and L. H. CHEN, *Chem. J. Chinese Universities*, In press.
- Y. SADAOKA, Y. SAKAI, M. MATSUMOTO and T. MANABE, J. Mater. Sci. 28 (1993) 5783.
- D. TRAN QUI, J. J. CAPPONI, M. GONDRAND, M. SAIB, J. C. JOUBERT and R. D. SHANNON, Solid State Ionics 3/4 (1981) 219.

Received 27 March and accepted 9 July 2003